

**MAGNETIC ANISOTROPY OF SOFT-UNDERLAYER INDUCED BY  
MAGNETRON FIELD**

**Related Applications**

This application claims priority from Provisional Application Serial No.  
5 60/221,458 filed July 25, 2000, entitled "Uniaxial Anisotropy of Soft-Underlayer  
Induced by Magnetron Field," the entire disclosure of which is hereby incorporated  
herein by reference. This application is also related to the application entitled  
"Magnetic Anisotropy of Soft-Underlayer Induced by Seedlayer," filed along with  
this application, the entire disclosure of which is hereby incorporated herein by  
10 reference.

**Field of Invention**

This invention relates to perpendicular recording media, such as thin film  
magnetic recording disks having perpendicular recording, and to a method of  
15 manufacturing the media. The invention has particular applicability to high areal  
density magnetic recording media exhibiting low noise.

**Background**

The increasing demands for higher areal recording density impose  
20 increasingly greater demands on thin film magnetic recording media in terms of  
remanent coercivity ( $H_r$ ), magnetic remanance ( $M_r$ ), coercivity squareness ( $S^*$ ),

medium noise, i.e., signal-to-medium noise ratio (SMNR), and narrow track recording performance. It is extremely difficult to produce a magnetic recording medium satisfying such demanding requirements.

The linear recording density can be increased by increasing the Hr of the magnetic recording medium, and by decreasing the medium noise, as by maintaining very fine magnetically non-coupled grains. Medium noise in thin films is a dominant factor restricting increased recording density of high-density magnetic hard disk drives, and is attributed primarily to inhomogeneous grain size and intergranular exchange coupling. Accordingly, in order to increase linear density, medium noise must be minimized by suitable microstructure control.

According to the domain theory, a magnetic material is composed of a number of submicroscopic regions called domains. Each domain contains parallel atomic moments and is always magnetized to saturation, but the directions of magnetization of different domains are not necessarily parallel. In the absence of an applied magnetic field, adjacent domains may be oriented randomly in any number of several directions, called the directions of easy magnetization, which depend on the geometry of the crystal. The resultant effect of all these various directions of magnetization may be zero, as is the case with an unmagnetized specimen. When a magnetic field is applied, the domains most nearly parallel to the direction of the applied field grow in size at the expense of the others. This is called boundary displacement of the domains or the domain growth. A further increase in magnetic

field causes more domains to rotate and align parallel to the applied field. When the material reaches the point of saturation magnetization, no further domain growth would take place on increasing the strength of the magnetic field.

The ease of magnetization or demagnetization of a magnetic material depends  
5 on the crystal structure, grain orientation, the state of strain, and the direction and strength of the magnetic field. The magnetization is most easily obtained along the easy axis of magnetization but most difficult along the hard axis of magnetization. A magnetic material is said to possess a magnetic anisotropy when easy and hard axes exist. On the other hand, a magnetic material is said to be isotropic when there are no  
10 easy or hard axes.

“Anisotropy energy” is the difference in energy of magnetization for these two extreme directions, namely, the easy axis of magnetization and the hard axis of magnetization. For example, a single crystal of iron, which is made up of a cubic array of iron atoms, tends to magnetize in the directions of the cube edges along  
15 which lie the easy axes of magnetization. A single crystal of iron requires about  $1.4 \times 10^5$  ergs/cm<sup>3</sup> (at room temperature) to move magnetization into the hard axis of magnetization, which is along a cubic body diagonal.

The anisotropy energy  $U_A$  could be expressed in an ascending power series of the direction cosines between the magnetization and the crystal axes. For cubic  
20 crystals, the lowest-order terms take the form of Equation (1),

$$U_A = K_1(\alpha_1^2\alpha_2^2 + \alpha_2^2\alpha_3^2 + \alpha_3^2\alpha_1^2) + K_2(\alpha_1^2\alpha_2^2\alpha_3^2) \quad (1)$$

where  $\alpha_1$ ,  $\alpha_2$  and  $\alpha_3$  are direction cosines with respect to the cube, and  $K_1$  and  $K_2$  are temperature-dependent parameters characteristic of the material, called anisotropy constants.

Anisotropy constants can be determined from (1) analysis of magnetization curves, (2) the torque on single crystals in a large applied field, and (3) single crystal magnetic resonance.

The total energy of a magnetic substance depends upon the state of strain in the magnetic material and the direction of magnetization through three contributions. The first two consist of the crystalline anisotropy energy of the unstrained lattice plus a correction that takes into account the dependence of the anisotropy energy on the state of strain. The third contribution is that of the elastic energy, which is independent of magnetization direction and is a minimum in the unstrained state. The state of strain of the crystal will be that which makes the sum of the three contributions of the energy a minimum. The result is that, when magnetized, the lattice is always distorted from the unstrained state, unless there is no anisotropy.

“Magnetostriction” refers to the changes in dimension of a magnetic material when it is placed in magnetic field. It is caused by the rotation of domains of a magnetic material under the action of magnetic field. The rotation of domains gives rise to internal strains in the material, causing its contraction or expansion.

The requirements for high areal density impose increasingly greater requirements on magnetic recording media in terms of coercivity, remanent

squareness, low medium noise and narrow track recording performance. It is extremely difficult to produce a magnetic recording medium satisfying such demanding requirements, particularly a high-density magnetic rigid disk medium for longitudinal and perpendicular recording. The magnetic anisotropy of longitudinal and perpendicular recording media makes the easily magnetized direction of the media located in the film plane and perpendicular to the film plane, respectively. The remanent magnetic moment of the magnetic media after magnetic recording or writing of longitudinal and perpendicular media is located in the film plane and perpendicular to the film plane, respectively.

A substrate material conventionally employed in producing magnetic recording rigid disks comprises an aluminum-magnesium (Al-Mg) alloy. Such Al-Mg alloys are typically electrolessly plated with a layer of NiP at a thickness of about 15 microns to increase the hardness of the substrates, thereby providing a suitable surface for polishing to provide the requisite surface roughness or texture.

Other substrate materials have been employed, such as glass, e.g., an amorphous glass, glass-ceramic material which comprises a mixture of amorphous and crystalline materials, and ceramic materials. Glass-ceramic materials do not normally exhibit a crystalline surface. Glasses and glass-ceramics generally exhibit high resistance to shocks.

Figure 1 shows the schematic arrangement of a magnetic disk drive using a rotary actuator. A disk or medium 11 is mounted on a spindle 12 and rotated at a

predetermined speed. The rotary actuator comprises an arm 15 to which is coupled a suspension 14. A magnetic head 13 is mounted at the distal end of the suspension 14. The magnetic head 13 is brought into contact with the recording/reproduction surface of the disk 11. The rotary actuator could have several suspensions and multiple  
5 magnetic heads to allow for simultaneous recording and reproduction on and from both surfaces of each medium.

An electromagnetic converting portion (not shown) for recording/reproducing information is mounted on the magnetic head 13. The arm 15 has a bobbin portion for holding a driving coil (not shown). A voice coil motor 19 as a kind of linear  
10 motor is provided to the other end of the arm 15. The voice motor 19 has the driving coil wound on the bobbin portion of the arm 15 and a magnetic circuit (not shown). The magnetic circuit comprises a permanent magnet and a counter yoke. The magnetic circuit opposes the driving coil to sandwich it. The arm 15 is swingably supported by ball bearings (not shown) provided at the upper and lower portions of a  
15 pivot portion 17. The ball bearings provided around the pivot portion 17 are held by a carriage portion (not shown).

A magnetic head support mechanism is controlled by a positioning servo driving system. The positioning servo driving system comprises a feedback control circuit having a head position detection sensor (not shown), a power supply (not  
20 shown), and a controller (not shown). When a signal is supplied from the controller to the respective power supplies based on the detection result of the position of the

magnetic head 13, the driving coil of the voice coil motor 19 and the piezoelectric element (not shown) of the head portion are driven.

A cross sectional view of a conventional longitudinal recording disk medium is depicted in Figure 2. A longitudinal recording medium typically comprises a non-magnetic substrate 20 having sequentially deposited on each side thereof an underlayer 21, 21', such as chromium (Cr) or Cr-alloy, a magnetic layer 22, 22', typically comprising a cobalt (Co) -base alloy, and a protective overcoat 23, 23', typically containing carbon. Conventional practices also comprise bonding a lubricant topcoat (not shown) to the protective overcoat. Underlayer 21, 21', magnetic layer 22, 22', and protective overcoat 23, 23', are typically deposited by sputtering techniques. The Co-base alloy magnetic layer deposited by conventional techniques normally comprises polycrystallites epitaxially grown on the polycrystal Cr or Cr-alloy underlayer.

The underlayer and magnetic layer are conventionally sequentially sputter deposited on the substrate in an inert gas atmosphere, such as an atmosphere of pure argon. A conventional carbon overcoat is typically deposited in argon with nitrogen, hydrogen or ethylene. Conventional lubricant topcoats are typically about 20Å thick.

It is recognized that the magnetic properties, such as Hr, Mr, S\* and SMNR, which are critical to the performance of a magnetic alloy film, depend primarily upon the microstructure of the magnetic layer which, in turn, is influenced by one or more underlying layers on which it is deposited. It is also recognized that an underlayer

made of soft magnetic films is useful in perpendicular recording media because a relatively thick (compared to magnetic layer) soft underlayer provides a return path for the read-write head and amplifies perpendicular component of the write field in the recording layer. However, Barkhausen noise caused by domain wall motions in the soft underlayer can be a significant noise source. Since the orientation of the domains can be controlled by the magnetic anisotropy, introducing a magnetic anisotropy in the soft underlayer would be one way to suppress Barkhausen noise. When the magnetic anisotropy is sufficiently large, the domains would preferably orient themselves along the anisotropy axis.

The magnetic anisotropy could be controlled in several ways in the soft magnetic thin film materials. The most frequently applied methods are post-deposition annealing while applying a magnetic field and applying a bias magnetic field during deposition. However, both methods can cause complications in the disk manufacturing process.

A "soft magnetic" material is material that is easily magnetized and demagnetized. As compared to a soft magnetic material, a "hard magnetic" material is one that neither magnetizes nor demagnetizes easily. The problem of making soft magnetic materials conventionally is that they usually have many crystalline boundaries and crystal grains oriented in many directions. In such metals, the magnetization process is accompanied by much irreversible Bloch wall motion and by much rotation against anisotropy, which is usually irreversible. See *Mc-Graw Hill*



*Encyclopedia of Science & Technology*, Vol. 5, 366 (1982). *Mc-Graw Hill*

*Encyclopedia of Science & Technology* further states that the preferred soft material would be a material fabricated by some inexpensive technique that results in all crystal grains being oriented in the same or nearly the same direction. *Id.*

- 5 Applicants, however, have found that “all grains” oriented in the same direction would be very difficult to produce and would not be the “preferred soft material.” In fact, applicants have found that very high anisotropy is not desirable.

This invention describes how one can create magnetic anisotropy in soft underlayer by maximizing the effect of a magnetron field and minimizing  
10 magnetostriction effect.

### **Summary of the Invention**

The invention provides a perpendicular magnetic recording medium having high areal recording density exhibiting low noise. One way of achieving this goal is  
15 to produce a magnetic film in the perpendicular magnetic recording medium with an easy axis substantially directed in a transverse direction to a traveling direction of a read-write head.

One embodiment is a recording device for perpendicular recording comprising a magnetic head and a recording medium, the recording medium  
20 comprising a substrate and a magnetic underlayer on the substrate, the underlayer comprising an easy axis of magnetization directed in a direction substantially

transverse to a traveling direction of the magnetic head. The underlayer could comprise a substantially radial or transverse anisotropy.

The underlayer could comprise a soft magnetic material. The underlayer provides a return path for a recording head. The underlayer could also amplify a perpendicular component of a write field in a recording layer overlying the underlayer. Preferably, the underlayer would have low magnetostriction about 0, typically lower than  $1-5 \times 10^{-6}$ . The underlayer could comprise a material selected from the group consisting of a permalloy, a CoZrNb alloy, a NiFe alloy and a FeAlN alloy. The recording medium could be a disk, a tape or any other device capable of recording data.

In one embodiment, the underlayer has the easy axis of magnetization induced by a magnetron field and the thickness of the underlayer could be about 200-400 nm.

Another embodiment is a method for manufacturing a magnetic recording disk for perpendicular recording, comprising applying a magnetron field and depositing an underlayer on a substrate, wherein the underlayer could comprise an easy axis of magnetization directed in a radial direction of the magnetic recording disk. The method could further comprise heating the substrate. The step of depositing an underlayer could be by sputtering, wherein the sputtering could be a reactive sputtering. In one variation, the substrate is kept stationary during the depositing a magnetic underlayer, wherein a diameter of a magnetron source producing the magnetron field is larger than a diameter of the substrate. In another

variation, the substrate is rotated during the depositing a magnetic underlayer, wherein a size of a magnetron source producing the magnetron field is smaller or comparable to a diameter of the substrate and the substrate is placed off-center with respect to the magnetron source.

5           Another embodiment is a disk drive comprising a magnetic recording disk for perpendicular recording, wherein the magnetic recording disk comprises a substrate and a magnetic underlayer on the substrate, wherein the underlayer comprises an easy axis of magnetization directed in a radial direction of the magnetic recording disk.

Yet another embodiment is a magnetic recording disk for perpendicular  
10   recording, comprising a substrate and means for providing a return path for a recording head. In this invention, means for providing a return path for a recording head comprises a soft magnetic layer having an easy axis of magnetization directed in a radial direction of the magnetic recording disk or directed in a transverse direction of a traveling direction of a magnetic head traveling over a tape or disk during a read-  
15   write operation.

As will be realized, this invention is capable of other and different embodiments, and its details are capable of modifications in various obvious respects, all without departing from this invention. Accordingly, the drawings and description are to be regarded as illustrative in nature and not as restrictive.

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#### **Brief Description of the Drawings**

Figure 1 is a view of a magnetic disk drive.

Figure 2 is a schematic representation of the film structure in accordance with a magnetic recording medium of the prior art.

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Figure 3 is perspective view of a magnetic head and a magnetic disk of a perpendicular recording disk medium.

Figures 4a and 4b are schematic diagrams of a magnetron source and disk substrate in embodiments of this invention showing a fixed magnetron source centered and off-centered from the center of a rotating disk substrate.

Figure 5 is a schematic representation of the film structure in accordance with a magnetic recording medium of this invention.

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Figure 6 shows hysteresis loops of an as-deposited  $\text{CoZr}_5\text{Nb}_4$  film measured in radial and circumferential directions using a DMS vector VSM.

Figure 7 shows the hysteresis loops measured in radial and circumferential directions of a  $\text{FeAlN}$  film.

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### Detailed Description

A perpendicular recording disk medium, shown in Figure 3, is similar to the longitudinal recording medium depicted in Figure 2, but with the following differences. First, a perpendicular recording disk medium has soft magnetic underlayer 31 of an alloy such as Permalloy instead of a Cr-containing underlayer. Second, as shown in Figure 3, magnetic layer 32 of the perpendicular recording disk medium comprises domains oriented in a direction perpendicular to the plane of the substrate 30. Also, shown in Figure 3 are the following: (a) read-write head 33 located on the recording medium, (b) traveling direction 34 of head 33 and (c) transverse direction 35 with respect to the traveling direction 34.

This invention provides magnetic recording media suitable for high areal recording density exhibiting high SMNR. This invention achieves such technological advantages by forming a substantially radial anisotropy in a soft underlayer. The underlayer is "soft" because it made of a soft magnetic material and it is called an "underlayer" because it resides under a recording layer. A seedlayer, on the other hand, is a layer lying in between the substrate and the underlayer.

In a preferred embodiment, a magnetron field produces the radial anisotropy in the soft underlayer. In a magnetron, electrons generated from a heated cathode move under the combined force of a radial electric field and an axial magnetic field. By its structure, a magnetron causes moving electrons to interact synchronously with



magnetic material of the underlayer are directed more toward a direction transverse to the traveling direction of the read-write head than toward the traveling direction.

Typically, when a magnetic recording medium is a tape, the tape travels and the head is stationary. Therefore, "a traveling direction of the magnetic head" of a stationary head of a recording device in which the magnetic recording tape moves is the direction in which the head "travels" spatially with respect to the magnetic recording tape.

In one embodiment of this invention, fixed magnetron sources with inner-and-outer magnet poles are used in a single disk type sputtering machine. Shunts are placed in the gap between the inner and outer magnet poles. Soft magnetic films with low magnetostriction are sputtered using the above magnetron sources on super-smooth disk substrates. Low magnetostriction is obtained by choosing either proper material such as permalloy and CoZrNb or proper sputtering conditions for the materials with higher magnetostriction such as NiFe55 and FeAlN. The disk substrates are kept either stationary (Example I) or rotating (Example II), depending upon the relative size of the magnetron source and the disk substrate. In Example I, the diameter of the source is larger than that of disks, and disks are concentric with the source during sputtering. In Example II, the size of magnetron source is smaller or comparable to that of the disk, and vertical distance between the source and substrate is relatively small.

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A disk substrate 41 is placed centered (Figure 4a) and off-centered (Figure 4b) with a target 42, to minimize stress in the films. In Figure 4a, the magnet array in the circular magnetron is symmetrical and the disk and magnetron are in concentric geometry. Therefore, no relative motion between magnetron and disk substrate is required. In another variation of the method shown in Figure 4a, the magnet array in circular magnetron can be asymmetrical. Thus either the disk or magnetron should be rotated to provide a radial field to the disk. However, as shown in Figure 4b, even if the disk is off-centered from the magnetron, a net field can be applied in the radial direction of the disk when the disk is rotated. Offset should be adjusted to obtain a radial field on the whole disk, depending upon size of disk, magnetron pack and magnetron field profile. The soft-underlayers manufactured according to the current methods had radial magnetic anisotropy induced by a field.

The term "induced" is used because magnetron field is external factor that causes anisotropy. Unlike magnetic anisotropy caused by magnetocrystalline or shape anisotropy, anisotropy formed by magnetron field is considered as induced anisotropy.

Applicants recognized that the soft-underlayers manufactured according to the current methods have radial magnetic anisotropy induced by a field by considering all possible origins of magnetic anisotropy:

(1) Magnetocrystalline anisotropy: Sputtered thin films listed as examples in this application form poly-crystalline microstructure. Crystallographic orientation of



micro-crystals in the films is random in the plane of disk. There is no reason why they form differently in radial and circumferential direction. Therefore, the disks of this invention do not have magnetocrystalline anisotropy.

(2) Shape anisotropy: This also does not explain the anisotropy between radial  
5 and circumferential direction of the disks of this invention.

(3) Stress anisotropy: Magnetostriction can cause stress-induced anisotropy. However, the disks of this invention had very low magnetostriction effect and, therefore, there could be substantially no stress anisotropy in the disks of this invention.

10 (4) External field induced anisotropy: When an external magnetic field is applied during film deposition, it can create short-range ordering that produces magnetic anisotropy in the direction of applied field. This method is used in manufacturing a read-write head of a recording medium. Similarly, this invention utilizes a magnetron field to apply a field in the radial direction of disks to produce  
15 radial anisotropy.

In accordance with embodiments of this invention, the substrates that may be used in the invention include glass, glass-ceramic, NiP/aluminum, metal alloys, plastic/polymer material, ceramic, glass-polymer, composite materials or other non-magnetic materials.

20 A preferred embodiment of a perpendicular recording medium of this invention is shown in Figure 5. The thickness of soft magnetic underlayer 52 is about

200-400 nm, and the thickness of magnetic layer 55 deposited on the underlayer is about 20 nm. In between the soft magnetic underlayer 52 and the magnetic layer 55 could be intermediate layers 53 and 54 of thickness of about 5-10 nm. Protective layer 56 typically covers the magnetic layer 55.

5           A perpendicular recording medium of this invention comprises a soft underlayer and a recording layer. A soft underlayer should preferably be made of soft magnetic materials and the recording layer should preferably be made of hard magnetic materials. A soft underlayer is relatively thick compared to other layers. Any layers between the soft underlayer and the recording layer is called interlayer or  
10   intermediate layer. An interlayer can be made of more than one layer of non-magnetic materials. The purpose of the interlayer is to prevent an interaction between the soft magnetic underlayer and recording layer. An interlayer could also promote the desired properties of the recording layer. Conventional (longitudinal) media do not have a soft magnetic underlayer. Therefore, the layers named as “underlayer,”  
15   “seed layer,” “sub-seed layer,” or “buffer layer” of longitudinal media are somewhat equivalent to the intermediate layer(s) of perpendicular media.

          “Anisotropy” in the examples below was determined as follows. First, the hysteresis loops of the soft underlayer material were measured along both the radial and circumferential directions of the magnetic recording disk. For example, Figures  
20   6 and 7 show representative hysteresis loops of the two soft underlayer materials of Examples 1 and 2, discussed below. From Figure 6 one can see that external field of

about 1 Oe can saturate the underlayer used in the disk of example of Figure 6 when a field is applied in the radial direction, while about 20 Oe is necessary to saturate the sample in circumferential direction. In other words, it is easier to saturate the sample in radial direction than in circumferential direction. Thus, radial and circumferential  
5 directions are called the easy and hard axis, respectively.

The underlayers of the disks whose hysteresis loops are shown in Figure 6 also have radial anisotropy. If there would be no anisotropy, the hysteresis loops for the radial and circumferential directions would superimpose on each other. The fact that these two hysteresis loops do not superimpose over one another indicates that the  
10 magnetic underlayer material has anisotropy. With reference to the hysteresis loops of Figure 6, "anisotropy" is determined as follows.

- (1) A straight line that goes through the origin and represents the slope of the initial portion of the hard axis hysteresis loop passing through or near the origin is drawn.
- 15 (2) The saturated part of the easy axis hysteresis loop is extended until it intersects with the line drawn in step (1).
- (3) The field that corresponds to the intersection of the two lines drawn in steps (1) and (2) is called  $H_k$  and it is a measure of the radial anisotropy of the soft magnetic underlayer sample.

In particular, the soft magnetic underlayers of the disks whose hysteresis loops are shown in Figure 6 and 7 have radial anisotropy values of 17 Oe and 30 Oe, respectively.

5           **Examples**

All samples described in this disclosure were fabricated with DC magnetron sputtering except carbon films were made with AC magnetron sputtering.

**Example I**

10           A single-disk type of sputtering machine with multi-vacuum chambers was used for fabricating CoZr5Nb4 alloy films as shown in Figure 4a. The diameter of magnetron source and disk were 7 inch and 84 mm, respectively. The films were sputtered on either heated or unheated substrates in argon gas pressure of 3 to 6 mtorr using 1 to 4 kW DC power. The hysteresis loops of an as-deposited CoZr5Nb4 film  
15           measured in radial and circumferential direction using a DMS vector VSM are shown in Figure 6. A "DMS vector VSM" is Model 10 VSM provided by Digital Measurement Systems. It is called vector VSM because the direction of field can be changed arbitrary with respect to sample.

          The particular film of the disk whose hysteresis loop is shown in Figure 6 was  
20           sputtered on unheated disk substrate by applying 4 kW power at argon pressure of 3 mtorr. Hysteresis loops of the films sputtered in different conditions showed similar

characteristics as shown in Figure 6, due to low magnetostriction, which is insensitive to deposition conditions. The term "different conditions" means different sputtering conditions such as different power and pressure, i.e., a wide sputtering process window is available to obtain CoZrNb films with low magnetostriction and thus obtain radial anisotropy induced by magnetron field. The term "similar characteristics" means that the soft underlayer has very low coercivity (1 Oe or less) in both radial and circumferential directions and it has a radial magnetic anisotropy of order of 15 Oe. As-deposited CoZrNb<sub>4</sub> films sputtered in any conditions had radial anisotropy induced by a magnetron field.

10            This method could be used with NiFe alloys. Due to different dependency of magnetostriction on the composition of the alloy, proper sputtering conditions that allow a magnetron field to take a full effect on inducing anisotropy are different for different compositions. For example, NiFe<sub>25</sub> films that had almost zero magnetostriction showed the strongest radial anisotropy when they were sputtered on an unheated substrate with relatively low power, whereas NiFe<sub>40</sub> films were radially anisotropic when they were sputtered on heated substrates.

### **Example II**

20            A single-disk type of sputtering machine containing a multi-magnetron source was used for fabricating FeAlN films as shown in Figure 4b. The diameter of the magnetron source was 3 inch, and the diameter of disk was 84 mm or larger. The

films were sputtered on unheated substrates in the mixture of argon and nitrogen gas by reactive sputtering using FeAl alloy target. Nitrogen gas flow % and total gas pressure was varied from 0 to 15 % and 3 to 15 mtorr, respectively. The DC power was fixed at 400 W and offset was varied from about 0.25 to 1.55 inch. Here, offset  
5 is defined as the horizontal distance between the center of the disk and the center of the source. The disk substrates were rotated in 50 rpm. The orientation of magnetic anisotropy and the magnitude of the anisotropy field of the films showed very complicated dependency on nitrogen %, total gas pressure, and thickness. The films sputtered at large offset with low nitrogen flow % showed radial anisotropy induced  
10 by magnetron field. Figure 7 shows the hysteresis loops measured in radial and circumferential directions for a 400 nm thick film sputtered at 1.55 inch offset. The film was sputtered at 2% nitrogen flow and total gas pressure of 3 mtorr, and showed radial anisotropy.

By this invention, radial magnetic anisotropy can be induced in soft  
15 underlayer films deposited on disk substrates by using a fixed pole DC-magnetron sputtering source. Magnetostriction effects on the anisotropy need to be minimized by choosing materials with low magnetostriction, or adjusting sputtering conditions.

The above description is presented to enable a person skilled in the art to make and use the invention, and is provided in the context of a particular application  
20 and its requirements. Various modifications to the preferred embodiments will be readily apparent to those skilled in the art, and the generic principles defined herein

may be applied to other embodiments and applications without departing from the spirit and scope of the invention. Thus, this invention is not intended to be limited to the embodiments shown, but is to be accorded the widest scope consistent with the principles and features disclosed herein.

5           This application discloses several numerical range limitations. Persons skilled in the art would recognize that the numerical ranges disclosed inherently support any range within the disclosed numerical ranges even though a precise range limitation is not stated verbatim in the specification because this invention can be practiced throughout the disclosed numerical ranges. A holding to the contrary would “let form  
10 triumph over substance” and allow the written description requirement to eviscerate claims that might be narrowed during prosecution simply because the applicants broadly disclose in this application but then might narrow their claims during prosecution. Finally, the entire disclosure of the patents and publications referred in this application are hereby incorporated herein by reference.

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